# Bismuth Tellurides: $\mathbf{B i T e}$ and $\mathrm{Bi}_{\mathbf{4}} \mathbf{T e}_{\mathbf{3}}$ 

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(Received 13 April 1978; accepted 12 September 1978)


#### Abstract

BiTe, trigonal, $P \overline{3} m 1, a=4.423$ (2), $c=$ 24.002 (6) Å, $Z=6, D_{x}=4.966 \mathrm{Mg} \mathrm{m}^{-3}, \mu r($ Mo Ka) $=4.2 ; \mathrm{Bi}_{4} \mathrm{Te}_{3}$, rhombohedral, $R \overline{3} m, a_{\text {hex }}=4.451$ (1), $c_{\text {hex }}=41.888(5) \AA, Z_{\text {hex }}=3, D_{x}=5.087 \mathrm{Mg} \mathrm{m}^{-3}$, $\mu r($ Mo Ka $)=5.9$. Both structures are characterized by the combination of 5 -layers ( TeBiTeBiTe ) and 2-layers (BiBi). The unit cell of BiTe contains 12 atomic layers stacked along the $c$ axis in the order $\mathrm{Te}(1) \mathrm{Bi}(1) \mathrm{Te}(2)-$ $\mathrm{Bi}(2) \mathrm{Te}(3) \mathrm{Bi}(3) \mathrm{Bi}(3) \mathrm{Te}(3) \mathrm{Bi}(2) \mathrm{Te}(2) \mathrm{Bi}(1) \mathrm{Te}$ (1). Interatomic distances between the nearest pairs are $3.044 \AA$ for $\mathrm{Te}(1)-\mathrm{Bi}(1), 3 \cdot 355 \AA$ for $\mathrm{Bi}(1)-\mathrm{Te}(2), 3 \cdot 137 \AA$ for $\mathrm{Te}(2)-\mathrm{Bi}(2), 3 \cdot 166 \AA$ for $\mathrm{Bi}(2)-\mathrm{Te}(3), 3 \cdot 326 \AA$ for $\mathrm{Te}(3)-\mathrm{Bi}(3), 3.267 \AA$ for $\mathrm{Bi}(3)-\mathrm{Bi}(3)$ and $3.679 \AA$ for $\mathrm{Te}(1)-\mathrm{Te}(1)$. One third of the unit cell of $\mathrm{Bi}_{4} \mathrm{Te}_{3}$ has seven layers in the order $\mathrm{Te}(1) \mathrm{Bi}(1) \mathrm{Te}(2) \mathrm{Bi}(1) \mathrm{Te}(1)$ $\mathrm{Bi}(2) \mathrm{Bi}(2)$ : the rhombohedral cell contains three such stacks. The interatomic distances are $3.327 \AA$ for $\mathrm{Te}(2)-\mathrm{Bi}(1), 3 \cdot 115 \AA$ for $\mathrm{Bi}(1)-\mathrm{Te}(1), 3 \cdot 401 \AA$ for $\mathrm{Te}(1)-\mathrm{Bi}(2)$ and $3 \cdot 098 \AA$ for $\mathrm{Bi}(2)-\mathrm{Bi}(2)$.


Introduction. Mixtures of Bi and Te ingots were sealed in evacuated silica tubes, with ratios corresponding to BiTe and $\mathrm{Bi}_{4} \mathrm{Te}_{3}$, respectively. They were melted at a temperature higher than 1073 K , and then cooled in liquid nitrogen. The solidified materials were ground to powders and again sealed into silica tubes under vacuum. After heating at a temperature higher than the melting point for a few days, the temperature was gradually lowered to 823 K in 3 d for BiTe , and to 733 K in 1 d for $\mathrm{Bi}_{4} \mathrm{Te}_{3}$. The tubes, which were held at those temperatures for 14 d for BiTe and 53 d for $\mathrm{Bi}_{4} \mathrm{Te}_{3}$, were then quenched in ice water.
Crystals used for recording diffraction patterns and intensity measurements were cut from aggregates of crystals. Both kinds of crystals were ground into spheres, and dipped into a mixture of $\mathrm{H}_{2} \mathrm{SO}_{4}, \mathrm{HNO}_{3}$ and HCl to eliminate powder lines on X-ray photographs due to grinding. Spheres with diameters of 0.13 and 0.15 mm were obtained for BiTe and $\mathrm{Bi}_{4} \mathrm{Te}_{3}$, respectively. Lattice constants and diffraction symbols for them correspond to those reported by Ozawa \&

[^0]Shimazaki (1974). Each of the synthesized products was confirmed to consist of a single phase.
The intensity data were collected on an automatic four-circle diffractometer (Philips PW 1100), up to $2 \theta$ $=70^{\circ}$ (graphite-monochromated Mo Ka radiation), adopting the $\omega-2 \theta$ scan mode. Scan speed was $5^{\circ}$ $\min ^{-1}$ in $\omega$ for BiTe and $3^{\circ} \min ^{-1}$ for $\mathrm{Bi}_{4} \mathrm{Te}_{3}$. The scan widths were determined according to the formula ( $a+$ $b \tan \theta)^{\circ}$, where $a=1 \cdot 0, b=0.5$ for BiTe and $a=1.5$, $b=0.8$ for $\mathrm{Bi}_{4} \mathrm{Te}_{3}$.

For BiTe , the intensity data were collected for 272 reflections satisfying the condition $I_{\text {top }}-2 I_{\text {top }}^{1 / 2} \geq I_{\mathrm{bck}}$, where $I_{\text {top }}$ is intensity in counts s ${ }^{-1}$ measured at the top of the reflection and $I_{\mathrm{bck}}$ is mean intensity in counts s ${ }^{-1}$ of both background measurements. For $\mathrm{Bi}_{4} \mathrm{Te}_{3}$, all the 463 reflections were measured. Lorentz, polarization and absorption corrections were applied to both sets of intensity data.

According to Imamov \& Semiletov (1971), crystal structures in the system $\mathrm{Bi}-\mathrm{Te}$ consist of 5 -layers (TeBiTeBiTe) and 2-layers (BiBi), which are stacked along the $c$ axis of a hexagonal lattice. For BiTe and $\mathrm{Bi}_{4} \mathrm{Te}_{3}$, they gave only the stacking sequences of the atomic layers: TeBiTeBiTeBiBiTeBiTeBiTe in the former and TeBiTeBiTeBiBi in the latter. These models were also confirmed by the Patterson-like method (Kihara \& Matsumoto, 1978). The structures of BiTe and $\mathrm{Bi}_{4} \mathrm{Te}_{3}$, were refined by the full-matrix least-squares program LINUS (Coppens \& Hamilton, 1970), starting from the models where each atomic layer is separated by an equal distance. Scattering factors for nonionized atoms and dispersion corrections were taken from International Tables for X-ray Crystallography

Table 1. The atomic coordinates with estimated standard deviations in parentheses

| (a) BiTe |  |  | (b) $\mathrm{Bi}_{4} \mathrm{Te}_{3}$ |  |  |  |  |
| :--- | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | $x$ | $y$ | $z$ |  | $x$ | $y$ | $z$ |
|  | $\mathrm{Te}(1)$ | $\frac{1}{3}$ | $\frac{2}{3}$ | $0.0552(3)$ | $\mathrm{Te}(2)$ | 0 | 0 |
| $\mathrm{Bi}(1)$ | 0 | 0 | $0.1242(3)$ | $\mathrm{Bi}(1)$ | $\frac{2}{3}$ | $\frac{1}{3}$ | $0.0505(1)$ |
| $\mathrm{Te}(2)$ | $\frac{2}{3}$ | $\frac{1}{3}$ | $0.2149(5)$ | $\mathrm{Te}(1)$ | $\frac{1}{3}$ | $\frac{2}{3}$ | $0.0927(2)$ |
| $\mathrm{Bi}(2)$ | $\frac{1}{3}$ | $\frac{2}{3}$ | $0.2908(3)$ | $\mathrm{Bi}(2)$ | 0 | 0 | $0.1460(1)$ |
| $\mathrm{Te}(3)$ | 0 | 0 | $0.3687(6)$ |  |  |  |  |
| $\mathrm{Bi}(3)$ | $\frac{2}{3}$ | $\frac{1}{3}$ | $0.4575(3)$ |  |  |  |  |

(1962). All the 272 measured reflections for BiTe were used in the course of the least-squares refinement. The 290 reflections with $|F|$ 's larger than $3 \sigma|F|$ were used in the refinement for $\mathrm{Bi}_{4} \mathrm{Te}_{3}$. Isotropic extinction parameters were introduced into these calculations. The final $R$ values of BiTe and $\mathrm{Bi}_{4} \mathrm{Te}_{3}$ were 0.064 and 0.068 , respectively. The final atomic coordinates are given in Table 1.* The interatomic distances and the bond angles of BiTe and $\mathrm{Bi}_{4} \mathrm{Te}_{3}$ are listed in Tables 2 and 3, respectively, with those of $\mathrm{Bi}_{2} \mathrm{Te}_{3}$ (Lange, 1939) for comparison.

Discussion. In the system $\mathrm{Bi}-\mathrm{Te}$, only the crystal structure of $\mathrm{Bi}_{2} \mathrm{Te}_{3}$ has been determined accurately (Lange, 1939; Nakajima, 1963). Recently, minerals with compositions BiTe and $\mathrm{Bi}_{4} \mathrm{Te}_{3}$ were found by Ozawa \& Shimazaki (1974). Both compositions are included in a range of the mineral wehrlite, $\mathrm{Bi}_{2+x} \mathrm{Te}_{3-x}$.

The structures of $\mathrm{Bi}_{2} \mathrm{Te}_{3}, \mathrm{BiTe}$ and $\mathrm{Bi}_{4} \mathrm{Te}_{3}$ are shown schematically in Fig. 1. They consist of layers of Bi and Te atoms perpendicular to the $c$ axis of a hexagonal lattice, where atoms are located on nearly cubic-closepacking positions.

The crystal structure of BiTe consists of twelve layers with six Bi and six Te atomic layers stacked in the order: $\mathrm{Te}(1) \mathrm{Bi}(1) \mathrm{Te}(2) \mathrm{Bi}(2) \mathrm{Te}(3) \mathrm{Bi}(3) \mathrm{Bi}(3) \mathrm{Te}(3)-$ $\mathrm{Bi}(2) \mathrm{Te}(2) \mathrm{Bi}(1) \mathrm{Te}(1)$. The order of atomic layers, $\mathrm{TeBi}-$ TeBiTe as seen in $\mathrm{Bi}_{2} \mathrm{Te}_{3}$, is also retained in this

[^1]Table 2. Bond lengths of $\mathrm{Bi}_{2} \mathrm{Te}_{3}, \mathrm{BiTe}$ and $\mathrm{Bi}_{4} \mathrm{Te}_{3}(\AA)$
Parallel values in the table correspond to the same kind of bond in both structures.

| $\mathrm{Bi}_{2} \mathrm{Te}_{3}{ }^{*}$ | BiTe | $\mathrm{Bi}_{4} \mathrm{Te}_{3}$ |
| :---: | :---: | :---: |
| $\mathrm{Te}(1)-\mathrm{Te}$ (1) 3.57 | $\mathrm{Te}(1)-\mathrm{Te}(1) 3.679$ (6) |  |
| $\mathrm{Te}(1)-\mathrm{Bi}(\mathrm{I}) 3 \cdot 10$ | $\mathrm{Te}(1)-\mathrm{Bi}(1) 3.044$ (5) |  |
| $\mathrm{Bi}(1)-\mathrm{Te}(2) 3.22$ | $\mathrm{Bi}(1)-\mathrm{Te}(2) 3.355$ (6) |  |
|  | $\mathrm{Te}(2)-\mathrm{Bi}(2) 3 \cdot 137$ (5) | $\mathrm{Te}(2)-\mathrm{Bi}(1) 3.327$ (2) |
|  | $\mathrm{Bi}(2)-\mathrm{Te}(3) 3 \cdot 166$ (9) | $\mathrm{Bi}(1)-\mathrm{Te}(1) 3 \cdot 115$ (5) |
|  | $\mathrm{Te}(3)-\mathrm{Bi}(3) 3.326$ (6) | $\mathrm{Te}(1)-\mathrm{Bi}(2) 3.401$ (6) |
|  | $\mathrm{Bi}(3)-\mathrm{Bi}(3) 3.267$ (6) | $\mathrm{Bi}(2)-\mathrm{Bi}(2) 3.098$ (4) |
|  | * Lange (1939). |  |

Table 3. Bond angles

| BiTe |  | $\mathrm{Bi}_{4} \mathrm{Te}_{3}$ |  |
| :--- | :--- | :--- | :--- |
| $\mathrm{Te}(1)-\mathrm{Te}(1)-\mathrm{Te}(1)$ | $73^{\circ} 54^{\prime}\left(11^{\prime}\right)$ | $\mathrm{Te}(2)-\mathrm{Bi}(1)-\mathrm{Te}(2)$ | $83^{\circ} 50^{\prime}\left(5^{\prime}\right)$ |
| $\mathrm{Te}(1)-\mathrm{Bi}(1)-\mathrm{Te}(1)$ | $93^{\circ} 10^{\prime}\left(9^{\prime}\right)$ | $\mathrm{Bi}(1)-\mathrm{Te}(1)-\mathrm{Bi}(1)$ | $91^{\circ} 03^{\prime}\left(11^{\prime}\right)$ |
| $\mathrm{Bi}(1)-\mathrm{Te}(2)-\mathrm{Bi}(1)$ | $82^{\circ} 47^{\prime}\left(15^{\prime}\right)$ | $\mathrm{Te}(1)-\mathrm{Bi}(2)-\mathrm{Te}(1)$ | $81^{\circ} 36^{\prime}\left(8^{\prime}\right)$ |
| $\mathrm{Te}(2)-\mathrm{Bi}(2)-\mathrm{Te}(2)$ | $89^{\circ} 40^{\prime}\left(14^{\prime}\right)$ | $\mathrm{Bi}(2)-\mathrm{Bi}(2)-\mathrm{Bi}(2)$ | $91^{\circ} 42^{\prime}\left(7^{\prime}\right)$ |
| $\mathrm{Bi}(2)-\mathrm{Te}(3)-\mathrm{Bi}(2)$ | $88^{\circ} 36^{\prime}\left(20^{\prime}\right)$ |  |  |
| $\mathrm{Te}(3)-\mathrm{Bi}(3)-\mathrm{Te}(3)$ | $83^{\circ} 24^{\prime}\left(15^{\prime}\right)$ |  |  |
| $\mathrm{Bi}(3)-\mathrm{Bi}(3)-\mathrm{Bi}(3)$ | $85^{\circ} 11^{\prime}\left(11^{\prime}\right)$ |  |  |



Fig. 1. Projections of structures of $\mathrm{Bi}_{2} \mathrm{Te}_{3}, \mathrm{BiTe}$ and $\mathrm{Bi}_{4} \mathrm{Te}_{3}$ on (110). 5- and 2-layer stacks are expressed by the numbers 5 and 2 , respectively.
structure. The 2 -layers composed of $\mathrm{Bi}(3)$ are sandwiched by the 5 -layers. Such a 2 -layer arrangement of Bi atoms is seen in the structure of bismuth. $\mathrm{Te}(1)$ is surrounded by six neighbors in adjacent layers, of which three are $\mathrm{Bi}(1)$ atoms at a distance of $3.044 \AA$ and the others are $\mathrm{Te}(1)$ atoms at a distance of 3.679 $\AA$. Whereas $\mathrm{Te}(1)-\mathrm{Te}(1)$ is the longest bond distance for neighboring atomic pairs, it is much shorter than the theoretically estimated van der Waals separation, $4.40 \AA$ (Von Hippel, 1948). Te(3) located on the other side of the 5 -layer is surrounded by six Bi atoms. The bond distance $\mathrm{Te}(3)-\mathrm{Bi}(3), 3.326 \AA$, is longer than $3.166 \AA$ for $\mathrm{Te}(3)-\mathrm{Bi}(2)$ and $3.267 \AA$ for $\mathrm{Bi}(3)-\mathrm{Bi}(3)$, but slightly shorter than the bond $\mathrm{Bi}(1)-\mathrm{Te}(2), 3.355$ $\AA$ in the 5 -layer. Thus, it is suggested that the bonding between the 5 - and 2-layer stacks is stronger than that between the 5 -layer stacks.

The crystal structure of $\mathrm{Bi}_{4} \mathrm{Te}_{3}$ is composed of atomic layers in the following order: $\mathrm{Te}(1) \mathrm{Bi}(1) \mathrm{Te}(2)-$ $\mathrm{Bi}(1) \mathrm{Te}(1) \mathrm{Bi}(2) \mathrm{Bi}(2)$. In this structure, the 5 -layer stacks and the 2-layer stacks are also retained and alternately stack along the $c$ axis. There are three 5 layer stacks and three 2-layer stacks in the unit cell. The 5 -layer stacks are centrosymmetrical with respect to the $\mathrm{Te}(2)$. The bond distance between the 5 - and 2 layer stack, $3.401 \AA$, is the longest for all neighboring atomic pairs and longer than the corresponding one in BiTe. Thus, this structure has the weakest bonding at the boundary between the 5 -and 2 -layer stacks.

As seen above, whereas the structures of $\mathrm{Bi}_{4} \mathrm{Te}_{3}$ and BiTe are both composed of the 5 - and the 2-layer stacks, the nature of the bonding within them is apparently affected by the stacking order of these stacks.

The authors wish to express their gratitude to Professor S. Sugiura for his encouragement throughout this study and to Mr J. B. Parise for improvement of the manuscript. The computations were carried out at the Computation Centers of this University (FACOM 230-35), and Tokyo University (HITAC 8700/8800).

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# Structure of Divalent-Europium Metaborate 

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(Received 17 July 1978; accepted 9 October 1978)


#### Abstract

EuB}_{2} \mathrm{O}_{4}\), orthorhombic, Pnca; $a=$ 6.593 (1), $b=12.063$ (2), $c=4.343$ (1) $\AA, Z=4$, $D_{x}=4.57, D_{m}=4.61 \mathrm{Mg} \mathrm{m}^{-3}, \mu\left(\right.$ Mo $\left.K_{(r)}\right)=18.15$ $\mathrm{mm}^{-1} . \mathrm{EuB}_{2} \mathrm{O}_{4}$ is isostructural with $\mathrm{CaB}_{2} \mathrm{O}_{4}$. The structural framework consists of endless chains of $\mathrm{BO}_{3}$ groups, $\left(\mathrm{BO}_{2}\right)_{\infty}$, along the $c$ axis. Each Eu atom is surrounded by eight O atoms to form an $\mathrm{EuO}_{8}$ dodecahedron. The mean Eu-Eu distances between Eu nearest neighbors and between Eu next-nearest neighbors are 4.080 and $6.632 \AA$, respectively. The final $R$ value was 0.029 for 334 observed reflections.


Introduction. Compounds containing divalent europium ( $\mathrm{Eu}^{2+}$ ) have been of interest because of their magnetic and spectroscopic properties. The magnetic exchange interactions in highly symmetrical compounds such as rock salt-type EuO or perovskite-type $\mathrm{EuTiO}_{3}$ have been evaluated on the basis of the $\mathrm{Eu}^{2+}-$ $\mathrm{Eu}^{2+}$ interactions, and $\mathrm{Eu}^{2+}-\mathrm{O}^{2-}-\mathrm{Eu}^{2+}$ angles, 90 and $180^{\circ}$, between Eu nearest neighbors and next-nearest neighbors respectively. These exchange interactions are sensitive to the distances between $\mathrm{Eu}^{2+}$ ions (Wolf, McGuire \& Shafer, 1964; McGuire, Shafer, Joenk, Alperin \& Pickart, 1966). Divalent-europium borates in the $\mathrm{EuO}-\mathrm{B}_{2} \mathrm{O}_{3}$ system, i.e. $\mathrm{EuB}_{4} \mathrm{O}_{7}, \mathrm{EuB}_{2} \mathrm{O}_{4}, \mathrm{Eu}_{2} \mathrm{~B}_{2} \mathrm{O}_{5}$ and $\mathrm{Eu}_{3} \mathrm{~B}_{2} \mathrm{O}_{6}$, have been synthesized and $\mathrm{EuB}_{2} \mathrm{O}_{4}$ is an antiferromagnet with the Néel temperature, $T_{N}=3 \mathrm{~K}$ (Hata, Adachi \& Shiokawa, 1977; Machida, Hata, Okuno, Adachi \& Shiokawa, 1978).

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0567-7408-79-010149-03\$01.00

The preparation of single crystals has been described in detail elsewhere (Machida et al., 1978). They were grown in the form of light-yellow needles. Weissenberg photographs exhibited the following systematic absences: $k+l=2 n+1$ for $0 k l, l=2 n+1$ for $h 0 l$ and $h=2 n+1$ for $h k 0$.

The intensity data were measured on a Rigaku automated four-circle diffractometer with a crystal of dimensions $0.15 \times 0.15 \times 0.30 \mathrm{~mm}$. Reflections within $(\sin \theta) / \lambda=0.71 \AA^{-1}$ were collected using the $\omega-2 \theta$ scan technique with Mo $K$ rr radiation ( $\lambda=0.7107 \AA$ ) monochromated by a graphite plate. 495 independent reflections were measured, among which 334 observed reflections were above background. The intensity data were corrected for Lorentz and polarization factors. No correction was made for absorption.

The location of the Eu atom was determined by a three-dimensional Patterson synthesis. Successive Fourier syntheses revealed the positions of the $O$ and $B$ atoms. All the atomic parameters were refined by the block-diagonal least-squares method with $H B L S$-V

Table 1. Final positional parameters Standard deviations are given in parentheses.

|  |  | $y$ | $z$ |
| :--- | :--- | :--- | :--- |
|  |  |  |  |
| Eu | $\frac{1}{4}$ | 0 | $0.2610(1)$ |
| B | $0.1247(13)$ | $0.1944(6)$ | $0.8281(18)$ |
| $\mathrm{O}(1)$ | $0.0935(9)$ | $0.0914(4)$ | $0.7350(9)$ |
| $\mathrm{O}(2)$ | $0.1440(9)$ | $0.2111(4)$ | $0.1470(12)$ |

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[^1]:    * Lists of structure factors, anisotropic temperature factors and the mean-square displacements have been deposited with the British Library Lending Division as Supplementary Publication No. SUP 33902 ( 8 pp.). Copies may be obtained through The Executive Secretary, International Union of Crystallography, 5 Abbey Square, Chester CHI 2HU, England.

